

Titanium Nitride Coatings on Inner Wall of Narrow Tube

by

by Yuta Kotani^{*1}, Koichi Hayakawa^{*1}, Tempei Tanakamaru^{*1} and Yoshihito Matsumura^{*2}

(Received on Mar. 31, 2008 and accepted on Jul. 9, 2008)

Abstract:

Titanium nitride (TiN) shows considerable hydrogen impermeability and is expected as a hydrogen-impermeable coating of cladding tubes for metal hydride fission fuel. The TiN coating production by the plasma process is very complicated because the cladding tube is not wide enough to create stable plasma. In this study, a simple reactive evaporation process was employed for the coating process of TiN films on the inner wall of a narrow tube. The films were thermally deposited on stainless steel sheets as the substrate and were coiled around the inner wall of silica glass tubes. A Ti wire twisted with a Mo filament was used for the evaporant. As a result, thin films with gold color were obtained. From the XRD measurement of the films, distinct diffraction patterns of TiN phase were observed. A reactivity of N₂ gas molecules with Ti atoms in Ti-N film formation was measured quantitatively. The reaction probability of N₂ molecules r_{N_2} is defined as the ratio of the N₂ absorption rate A_{N_2} on Ti to the rate of N₂ impingement Z_{N_2} . Reactive evaporation process showed $r_{N_2} \cong 0.013$. It was 1/5 of r_{N_2} with Ti films prepared by plasma processes

Keyword : Titanium nitride coating, Inner wall of narrow tube, Reactive vacuum evaporation process, prevention coating of hydrogen gas, reaction probability

1. Introduction

Narrow tubes are widely used in industry as water pipes, hydrogen gas pipes, cooling pipes for nuclear reactors, etc. However, when these steels are cathodically charged with hydrogen at room temperature, their subsurface layers become hard and brittle, and numerous cracks are formed. It is therefore necessary to protect the inside of the tubes. Titanium nitride shows considerable hydrogen impermeability and is expected as a prevention coating of hydrogen gas for a cladding tube of metal hydride fission fuel¹⁻⁵⁾. Among various methods of surface modification, plasma nitriding is widely used because it increases hardness and improves wear and fatigue resistance of ferrous and other materials⁶⁻⁹⁾. However, the coating processes are very complicated because the cladding tube is not wide enough to create stable plasma¹⁰⁾. In this study, a simple reactive evaporation process was employed for coating process of TiN films onto the inner wall of narrow tube. The reaction probability r_{N_2} of N₂ was used as a measure for assessing the reactivity and the reaction probability r_{N_2} of N₂ as a measure for assessing the reactivity for reactive vapor deposition process and plasma process¹¹⁾.

2. Experimental

The reactive evaporation process was used for Ti-N film formation. The schematic diagram of the evaporation apparatus used for this study is shown in Fig. 1.

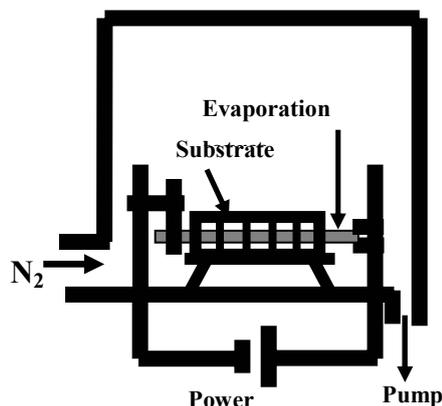


Fig. 1. Schematic of reactive evaporation process

The base pressure of the vacuum chamber was less than 3.0×10^{-4} Pa. The films were thermally deposited on stainless steel sheets (60 mm × 10 mm) as the substrate and were coiled around the inner wall of silica glass tubes (160 mm inner diameter and 10 mm long) was set nine pieces in series. The reactive evaporation system was used as the resistance heating process. A twisted Ti wire with a Mo

*1 Graduate Student, Course of Applied Science,
*2 Professor, Department of Energy Engineering,

filament was used for the evaporant. Nitrogen gas as reactive gas was introduced into the vacuum chamber with a gas pressure of 1.0×10^{-1} Pa. Deposition times were varied as Ti deposition time 300 s (5 min) and Ti-N deposition time 900 s (15 min) (sample Ti-N5/15), Ti deposition time 600 s (10 min) and Ti-N deposition time 600 s (10 min) (sample Ti-N10/10), Ti deposition time 600 s (10 min) and Ti-N deposition time 1200 s (20 min) (sample Ti-N10/20), respectively. After the film formation, film thickness was confirmed by an observation of film cross sections using a scanning electron microscope (SEM). The compositions of N/Ti in Ti-N films were analyzed by X-ray Photoelectron Spectroscopy (XPS). X-Ray diffraction (XRD) analyses was used for microstructure analysis.

3. Results and Discussion

3.1. Characteristics of Ti-N coatings

Deposited films showed golden color. The results on composition N/Ti and thickness conducted under different experimental Ti evaporation time and Ti-N evaporation times are presented in Table 1. Samples of Ti-N10/10 and Ti-N10/20 showed same thickness of deposition may imply that this film thickness is depended on the deposition time of Ti. The deposition rate of TiN was decreasing with scattering of Ti vapor by nitrogen gas molecules. After evaporation process, the filament showed yellowish gold color with titanium nitride formation. The reason for low deposition rate of TiN is also due to the decreasing vapor pressure of titanium with surface nitriding of titanium wire. The film composition of all films showed about $N/Ti = 0.6$. TiN phase shows $TiN/Ti = 0.56-1.0$ in Ti-N phase diagram. The composition of those film samples showed TiN by XPS analysis¹²⁾.

Table 1. Evaporation time vs. composition of Ti-N coatings

Sample	Ti evaporation time [s]	Ti-N evaporation time [s]	Composition N/Ti	Thickness [μm]
Ti-N5/15	300	900	0.65	2.5
Ti-N10/10	600	600	0.61	3.0
Ti-N10/20	600	1200	0.63	3.0

Fig. 2 shows the XRD diffraction patterns of Ti-N films as a function of various samples. Sample Ti-N5/15 shows strong

diffraction peak of titanium. And this sample shows weak titanium nitrides peaks and stainless steel as substrate peaks to penetrate film. Sample Ti-N10/20 shows strong diffraction peaks of TiN. And this sample shows strong stainless steel substrate peak to penetrate film and weak titanium peaks. Therefore, sample Ti-N10/20 is a mixed phase of TiN phase and Ti_2N phase. Diffraction peaks of Ti phase were observed as titanium under layer path through the thin TiN top coating.

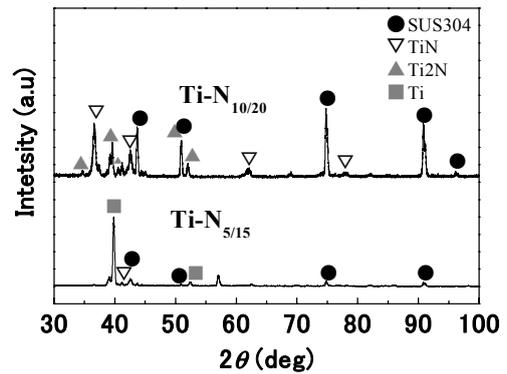


Fig. 2. X-ray diffraction profile of Ti-N thin films prepared at various evaporation times

3.2. Assessment of the reactivity

The reaction probability of a gas with a metal is a useful measure to assess the quantitative reactivity of gas involved in the reaction, which has been demonstrated for many metal-gas systems¹³⁻¹⁵⁾.

The total amount of Ti atoms in film N_{Ti} can be derived as follows.

$$N_{Ti} = (V_{Ti} \times \rho) / M_{Ti} \quad [\text{mol}]$$

Where V_{Ti} is volume of Ti film, ρ is Ti density and M_{Ti} is molecular mass of Ti.

The impinging rate of Ti atoms Z_{Ti} is defined as the ratio of the N_{Ti} times Avogadro's constant N_A to the evaporation time t :

$$Z_{Ti} = (N_{Ti} \times N_A) / t \quad [\text{s}^{-1}]$$

The absorbing rate of N_2 A_{N_2} on Ti surface can be derived as follows:

$$A_{N_2} = Z_{Ti} \times \{(N/2)/Ti\} \quad [\text{s}^{-1}]$$

The reaction probability of N_2 molecules r_{N_2} is defined as the ratio of the absorbing rate of N_2 on Ti surface A_{N_2} to the rate of N_2 impingement Z_{N_2} :

$$r_{N_2} = A_{N_2} / Z_{N_2} \quad (1)$$

where

$$Z_{N_2} = 2.6 \times 10^{24} P_{N_2} / (M_{N_2} \times T)^{1/2} [\text{m}^{-2} \cdot \text{s}^{-1}].$$

T is the absolute temperature, M is the molar mass and P_{N_2} is N_2 partial pressure.

Table 2. shows A_{N_2} of each sample.

Table 2. Absorbing rate of N_2 on Ti surface: A_{N_2} of samples

Sample	A_{N_2}
Ti-N _{5/15}	0.37×10^{20}
Ti-N _{10/10}	0.43×10^{20}
Ti-N _{10/20}	0.29×10^{20}

Table 3. shows the reaction probability r_{N_2} of samples. For $r_{N_2}=0.010-0.016$, almost same values are obtained. The r_{N_2} with Ti-N films by reactive evaporation process exhibit $r=0.013$.

Table 3. Reaction probability of samples

Sample	Reaction Probability r_{N_2}
Ti-N _{5/15}	0.013
Ti-N _{10/10}	0.016
Ti-N _{10/20}	0.010

Table 4 shows r_{N_2} on Ti surface for various processes ¹¹⁾. In Table 3 and Table 4, the reaction probability of N_2 with Ti film by reactive evaporation process and ARE process exhibit the same value of $r_{N_2}=0.07$.

Table 4. Reaction probability r_{N_2} on Ti surface for various processes

Process	Reaction Probability r_{N_2} at Composition $x=0.6$
ARE Process	0.07
HCD Process	0.15
UHV	0.25

In Table 4, the Ti surface prepared under UHV (Ultra-high vacuum, $P < 10^{-7}$ Pa) conditions exhibits the highest N_2 reactivity, $r_{N_2}=0.25$, even at $x=0.6$. Electron transfer between the surface and N_2 molecules is the essential step for the N_2 dissociation ¹⁶⁾. If the

surface is covered with residual gaseous (H_2O , CO_2 etc) and becomes less metallic, the electron transfer becomes inhibited. In this study, r_{N_2} with Ti film by reactive evaporation process exhibits 1/5 of r_{N_2} with Ti film prepared by plasma processes. This is because reactive evaporation process has a larger effect of residual gas than UHV ¹¹⁾.

4. Conclusion

In this study, a simple reactive evaporation process was employed for coating process of TiN films onto inner wall of narrow tube. The films were thermally deposited on stainless steel sheets as substrate. A twisted Ti wire with a Mo filament was used as evaporant. Thin films with gold color were obtained. A reactivity of N_2 gas molecules with Ti atoms in Ti-N film formation was measured quantitatively.

The r_{N_2} with Ti film by reactive evaporation process exhibits 1/4 of r_{N_2} with Ti film prepared by plasma process. Deposited Ti-N films showed golden color and composition of deposited films showed about N/Ti = 0.6 analyzed by XPS. Therefore, The simple reactive evaporation is very useful for TiN inner wall coating process of narrow tube.

Acknowledgement

This study was made in the frame of Development of Advanced Production Processes for Energy Conversion Materials, Future Science & Technology Joint Research Center, Tokai University.

References

- 1) K. Konashi and Y. Shimada, J. Nucl. Energy, vol. 46, (2004), pp. 457-466.
- 2) B. Tsuchiya, J. Huang, K. Konashi, M. Teshigawara and M. Yamawaki, Trans. Am. Nucl. Soc., vol. 81, (1999), pp. 125-126.
- 3) N. Hosoda, H. H. Uchida and E. Fromm, J. Less-Common. Met, vol. 172/174, (1991), pp. 824-831.
- 4) K. Ono, K. Saitoh, S. Inayashi and S. Tsukahara, J. Vac. Soc. Jpn, vol. 34, (1991), pp. 420-426.
- 5) T. Nishikiori, T. Nohira and Y. Ito, J. Electrochem. Soc, vol. 148, (2001), pp. E52-E59.
- 6) N. Nishida, H. Kamasaki, K. Hoda and N. Hosokawa, J. Met. Finishing Soc. Jpn, vol. 38, (1987), pp. 483-487.
- 7) P. A. Lindfors, W. M. Mularie and G. K. Wehner, Surf. Coating. Technol., vol. 29, (1986), pp. 275-290.
- 8) Y. Matsumura, T. Yoshioka and Y. C. Huang, Proc. Int. Ion Eng. Cong., Japan, 1983, pub. by Nippon Denki Gakkai, pp.1283.

- 9) A. K. Suri, R. Nimmagadda and R. F. Bunshah, *Thin Solid Films*, vol. 72, (1980), pp. 529-533.
- 10) T. Nagano and H. Fujiyama, *Jpn. J. Appl. Phys.*, vol. 38, (1999), pp. 4338-4341.
- 11) Y. Matsumura, T. Chujo and H. Uchida, *Surf. Coating Technol.*, vol. 60, (1993), pp. 489-492.
- 12) *Binary Alloy Phase Diagrams* second Edition, Hiroaki Okamoto, Thaddeus B. Massalski, P.R Subramanian and Linda Kacprzak. (Materials Park, Ohio 1990), pp. 2707.
- 13) S. Hofmann, *Thin Solid Films*, vol. 193-194, (1990), pp. 648-664.
- 14) Y. Matsumura, H. Uchida and Y. C. Huang, *Proc. Plasma Surf. Eng.*, Garmish-Partenkirchen, FRG, 1988, pub. by Deutsche Gesellschaft für Metallkunde, vol.1, pp.45.
- 15) E. From, V. Grajewski and H. H. Uchida, *Proc. Plasma Surface Engineering*, Garmish-Partenkirchen, FRG, 1998, pub. by Deutsche Gesellschaft für Metallkunde, vol.1, pp.37.
- 16) J. K. Norskov, *J. Less-Common Met.*, vol. 130, (1987), pp. 475-490.